

. Unclassified  CURITY CLASSIFICATION OF THIS PAGE (When Date Expedit	11 /-
REPORT DOCUMENTATION P. CE.	READ INSTRUCTIONS BEFORE COMPLETING FORM
	3. RECIPIENT'S CATALOG NUMBER
Technical Report No. 78-4	
TITLE (and Subtitle)	5. TYPE OF REPORT & PERIOD COVERED
Polyacetylene, (CH); n-Type and p-Type Doping and Compensation	Interim Technical Report
	6. PERFORMING ORG. REPORT NUMBER
AUTHOR(e)	S. CONTRACT OR GRANT NUMBER(+)
C. K. Chiang, S.C. Gau, C. R. Fincher, Jr.,	
Y. W. Park, A. G. MacDiarmid, and A. J. Heeger	NOO014-75-C-0962
PERFORMING ORGANIZATION NAME AND ADDRESS	10. PROGRAM ELEMENT, PROJECT, TASK AREA & WORK UNIT NUMBERS
Departments of Chemistry and Physics and the	
Matter, University of Pennsylvania, Philadelphia,	NR-356-602
CONTROLLING OFFICE NAME AND ADDRESS	12. REPORT DATE
Department of the Navy	August 4, 1978
Office of Naval Research	13. NUMBER OF PAGES
Arlington, Virginia 22217	14 15. SECURITY CLASS, (of this report)
. MONITORING AGENCY NAME & ADDRESS(II different from Controlling Office)	
	Unclassified
	15. DECLASSIFICATION/DOWNGRADING
	SCHEDULE
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DISTRIBUTION STATEMENT (of the abstract entered in Block 20, If different from	Report) U SEP 7 1978
	E
. SUPPLEMENTARY NOTES	
Published in Appl. Phys. Lett., 33, 18 (1978).	

9. KEY WORDS (Continue on reverse side if necessary and identify by block number)

Polyacetylene, (CH); n-type doping; p-type doping; compensation; donors; acceptors; p-n junctions; semiconductor device application; crystalline films; covalent polymer; electronic properties; doping techniques; Na-doped films; sodium naphthalides; electron transfer; ionic impurities; compensated polymer; charge transfer; polycation; polymer chains; band gap; bound states; electronic over

A series of experiments are reported which demonstrate that donors or acceptors can dope polyacetylene to n-type or p-type respectively, and that the two kinds of dopants can compensate one another. The formation of a rectifying p-n junction is demonstrated. The results suggest the possibility of utilizing doped polyacetylene in a variety of potential semiconductor device applications.

DD FORM 1473

EDITION OF 1 NOV 65 IS OBSOLETE S/N 0102-014-6601 | Unclassified

19. (Cont'd)

carriers; thermoelectric power; Seebeck coefficient; diode characteristic; I-V curve; optical studies; solar spectrum.

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OFFICE OF NAVAL RESEARCH

Task No. 356-602

TECHNICAL REPORT NO. 78-4

Polyacetylene, (CH) n-Type and p-Type Doping and Compensation

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N00014-75-C-0962

Polyacetylene, (CH): n-Type and p-Type Doping and Compensation

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#### Abstract

A series of experiments are reported which demonstrate that

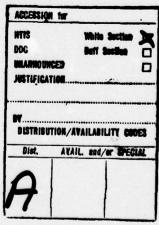
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suggest the possibility of utilizing doped polyacetylene in a variety of

potential semiconductor device applications.



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We have recently reported that when flexible, crystalline films of the covalent polymer, polyacetylene, (CH), are doped, their conductivity can be systematically and controllably varied over a range of more than eleven orders of magnitude. 1-3 The most highly conducting films exhibited a conductivity at room temperature of order 10° ohm-1 -cm-1, by far the highest observed for any covalent organic polymer. The electronic properties are similar to those of a simple classical semiconductor with thermally activated conductivity at low dopant levels and a transition to metallic properties at high levels. The dopant can be either an acceptor such as bromine, iodine or AsF, ; 1, 2 or the dopant can be a donor such as lithium, sodium and potassium. 3,4 We report in this paper on a series of experiments which demonstrate that both donors and acceptors can dope polyacetylene to n-type or p-type material respectively, and that the two kinds of dopants can compensate one another. The demonstration of successful n-type and p-type doping together with compensation indicates the possibility of utilizing doped polyacetylene in a variety of potential semiconductor device applications.

The (CH)<sub>x</sub> polymer films were prepared using refinements of the techniques developed by Shirakawa et al. <sup>5</sup> The doping techniques were developed recently in our laboratories. Details on the preparation and doping are published elsewhere. <sup>1,3,4</sup>

When pure polyacetylene is doped with an acceptor, the electrical conductivity increases sharply over many orders of magnitude at low

concentration, then saturates at higher dopant levels, above approximately 1%.  $^{2-4}$  The maximum conductivity we have reported to date at room temperature was  $560~\Omega^{-1}~\mathrm{cm}^{-1}$  for  $\mathrm{cis}[\mathrm{CH}(\mathrm{AsF_s})_{0.14}]_{\chi}^{2,3}$  Donor doping leads to qualitatively similar results. The highest conductivity obtained thus far through donor doping was  $200~\Omega^{-1}~\mathrm{cm}^{-1}$ . The typical behavior for the electrical conductivity as a function of dopant concentration is shown in Figure 1. The general features appear to be the same for the various donor and acceptor dopants, but with detailed differences in the saturation values and the critical concentration for the metal-insulator transition.

Earlier studies<sup>2,3</sup> suggested that compensation could be achieved.

The sensitivity of (CH)<sub>x</sub> to impurities implies that the residual conductivity in "pure" (CH)<sub>x</sub> is probably due to residual impurities or defects at very low levels. Exposure of pure (CH)<sub>x</sub> to vapor of the donor, NH<sub>3</sub>, causes the conductivity of trans-(CH)<sub>x</sub> to fall more than four orders of magnitude without detectable weight increase. The exposure to NH<sub>3</sub> thus results in compensation of the residual impurities and/or defects.

Compensation of n-type material by subsequent acceptor doping has been successfully demonstrated using Na (donor) and iodine or  $AsF_g$  (acceptors). Figures 2 and 3 show the compensation of Na-doped polyacetylene by iodine and  $AsF_g$ , respectively. The Na-doped films were prepared by treating the polymers with a solution of sodium naphthalide,  $^3$  Na $^+$ ( $C_{i \hat{o}}H_g$ ), in THF whereupon electron transfer from the naphthalide

radical anion to the (CH)<sub>x</sub> occurred. In each case the pure cis-polyacetylene was first doped with sodium until the conductivity was in the saturation range. Subsequent exposure to iodine or AsF<sub>s</sub> vapor resulted in the compensation curves plotted in Figs. 2 and 3. The compensation proceeds more slowly than the original doping; the electrical conductivity of the n-type sample gradually decreases and reaches a minimum. Continued doping with iodine or AsF<sub>s</sub> results in conversion to p-type material with an associated increase in conductivity.

Due to the relatively high concentration of ionic impurities in the sample, the conductivity of the fully compensated sample (at minimum) remained higher than that of the original polymer prior to doping. Nevertheless Figures 2 and 3 show compensation over approximately four orders of magnitude starting from the highest doping levels where the samples are metallic. Even more precise compensation can be expected using intermediate to low initial doping.

Starting with an initial composition (CHNa<sub>0.27</sub>)<sub>x</sub>, the compensation point with iodine occurred at (CHNa<sub>0.27</sub>I<sub>0.28</sub>)<sub>x</sub>; all compositions being determined by measurement of weight increase. Thus the compensation point corresponds approximately to a stoichiometric sodium to iodine ratio consistent (within the limits of error) to the presence of equal concentrations of Na<sup>+</sup> and I<sup>-</sup> in the compensated polymer. Continued doping leads to p-type material, where the iodine is known to be present as I<sub>3</sub><sup>-</sup> from Raman studies. 6, 7

From these observations, the overall shape of the compensation curves, and the fact that the doping can be removed chemically, we believe the doping results from charge transfer, and the compensation is best viewed as of direct chemical origin. Charge transfer occurs from polymer to acceptor (A) with the polymer chain acting as a polycation in the presence of an A species (in the case of iodine, the compensation and Raman experiments suggest I in the presence of excess Na going over to L in the p-type material). For a donor (D), the polymer chains act as a polyanion in the presence of the D species. The charge transfer need not be complete. The resulting electron (or hole) on the polymer chain is weakly bound to the D (or A ) ions by the Coulomb potential, forming donor (or acceptor) states in the gap. At low levels, thermal activation out of the bound states results in carriers for transport along the polymer chain. Above approximately 1 mole % doping, the bound states are screened, and the carriers are free to move along the polymer chains resulting in metallic behavior. 2 Compensation at least in the case of · iodine evidently occurs through formation of (D+A-) leaving the polymer chain essentially neutral and without electronic carriers.

The assignment of donor doped material as n-type and acceptor doped material as p-type follows from the chemical properties of the donor and acceptor dopants. Moreover, the assignments are consistent with the results obtained from studies of graphite intercalated with alkali

metals and iodine or AsF<sub>8</sub> respectively. <sup>8</sup> Finally thermoelectric power measurements on acceptor doped (CH)<sub>x</sub> yield a positive Seebeck coefficient consistent with p-type material. Moreover, the value of +15 µV/K found at room temperature for (CH)<sub>x</sub> heavily doped with AsF<sub>8</sub> is consistent with metallic behavior. Initial experiments directed toward fabrication of p-n junctions are encouraging as shown in Fig. 4. The junction was made by mechanically pressing together n-type (Na-doped) and p-type (AsF<sub>8</sub>-doped) strips of polymer film. Although some hysteresis is evident, a typical diode characteristic is seen in the I-V curve. Junctions have also been made using a single polymer strip doped n-type on one-half and p-type on the other half. Note that in all cases the forward bias direction was consistent with the p-type and n-type character of the acceptor and donor doped material.

In summary, polyacetylene exhibits many attractive features for potential device applications. From optical studies, 9 we conclude that the intrinsic band gap is at approximately 1.6 eV; i.e. reasonably well matched to the solar spectrum. In addition to the ability to dope to n-type or p-type and to compensate, the chemical and mechanical properties are attractive. The material can be synthesized in the form of strong, flexible, free-standing films or as films on glass or metal substrates.

Considering possible polyacetylene derivatives, replacement of some or all of the hydrogen atoms in (CH), with organic or inorganic

groups, copolymerization of acetylene with other acetylenes or olefins
and the use of different dopants, this large new class of organic polymer
semiconductors offer the possibility of unique chemical tuning of electrical
properties.

Acknowledgement: This work was supported by the Office of Naval Research.

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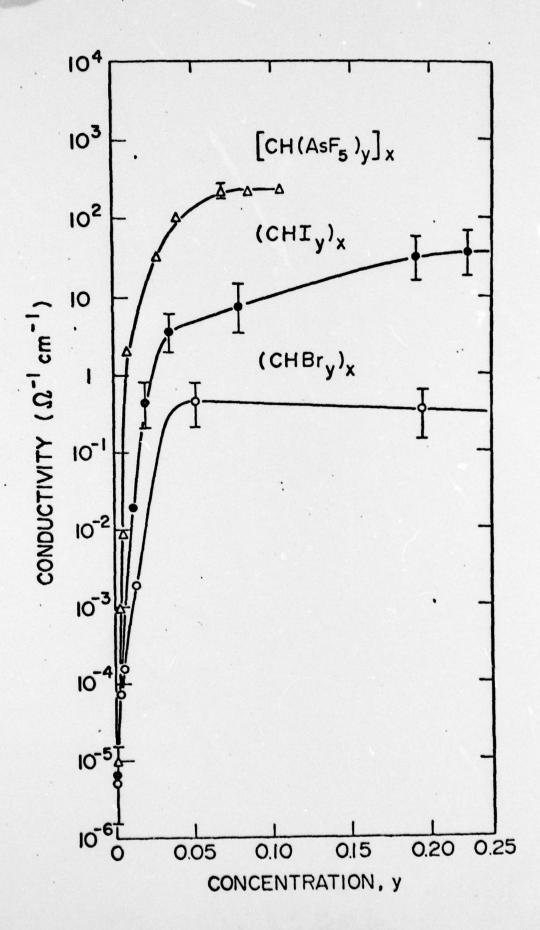
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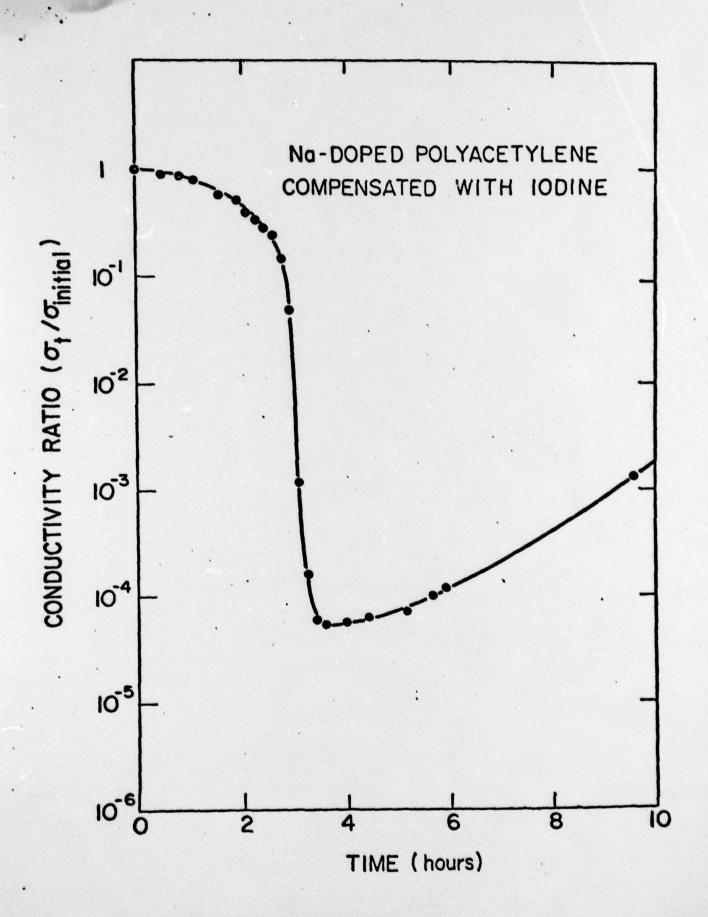
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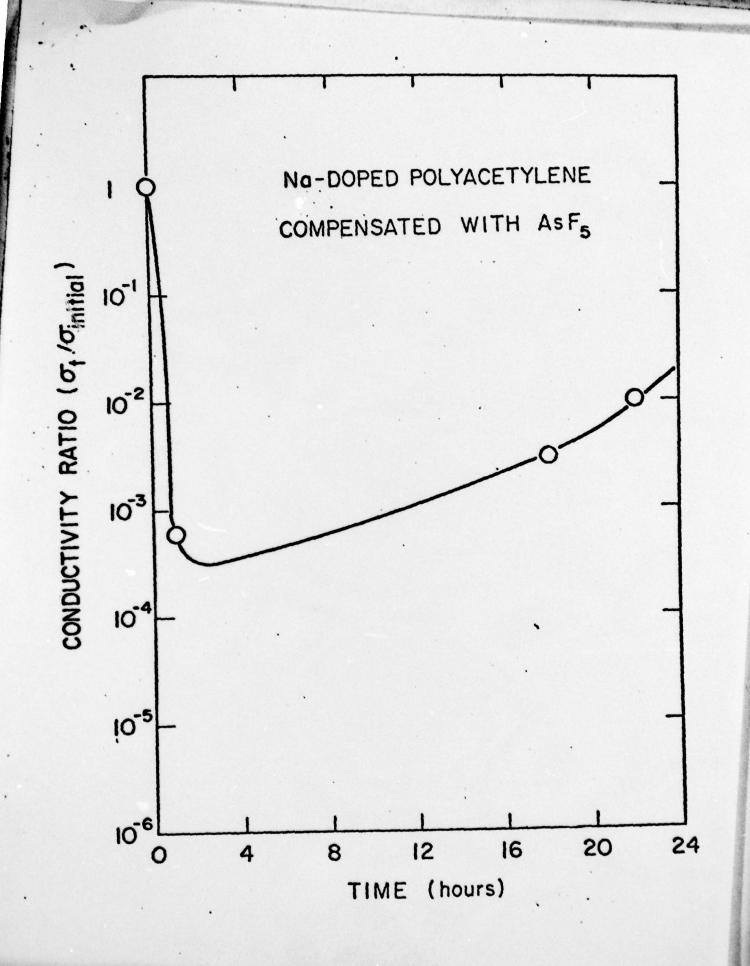
Figure 1: Electrical Conductivity (room temperature) as a function of dopant concentration:

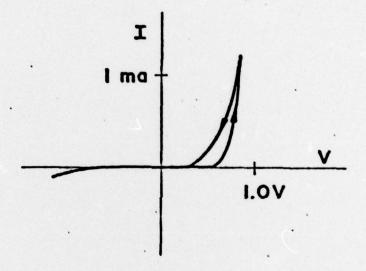
Bromine 000 Iodine ••• AsF<sub>g</sub>

- Figure 2: Compensation curve for Na-doped polyacetylene; conductivity ratio  $(\sigma_{(t)}/\sigma_{\text{initial}})$  vs. time. The sample was initially doped n-type and subsequently exposed to iodine vapor.
- Figure 3: Compensation curve for Na-doped polyacetylene; conductivity ratio  $(\sigma_{(t)}/\sigma_{\text{initial}})$  vs. time. The sample was initially doped n-type and subsequently exposed to AsF<sub>g</sub> vapor.
- Figure 4: I-V curve for a doped polyacetylene p-n junction.









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